# Contact mechanics and elements of tribology Lecture 5. Wear and Fretting

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# Outline

## 1 Wear

- Adhesive wear
- Abrasive wear
- Chemical wear
- Experimental testing

## 2 Fretting

- Basics
- Fretting wear
- Experimental determination of fretting crack initiation
- Numerical analysis of fretting initiation and propagation
- Fretting fatigue
- 3 Modelling fretting taking into account the contact gradient
- 4 Outlook : insight from numerical models

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Wear is a very general term which is defined as the degradation of the contact surface of a material in service. These degradations can be due to multiple factors, and it is now common to separate wear into 4 distinct families :

- adhesive wear;
- abrasive wear;
- chemical wear (corrosion);
- wear by contact fatigue.

We will first describe the first three types of wear, wear by contact fatigue will be discussed in more detail in section 2.

# Contents

## 1 Wear

#### Adhesive wear

- Abrasive wear
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# 2 Fretting

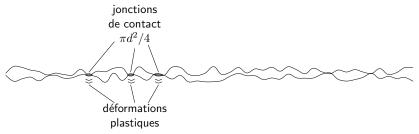
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Creation of strong bonds leading to the formation of junctions between the two materials at the level of asperities in contact.

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The adhesion index provides information on the propensity for two materials to adhere to each other if they are brought into contact.

It is expressed as a function of the effective Young's modulus  $E^*$ , the average radius of curvature of the asperities R, the standard deviation of the height distribution of the asperities  $\sigma$  and the adhesive work  $W_{ad}$ :

$$\alpha = \frac{E^*}{W_{ad}} \left(\frac{\sigma^3}{R}\right)^{1/2}$$

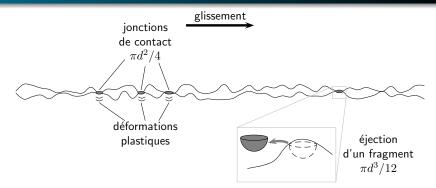
In practice, if  $\alpha \leq 5$ , risks of adhesion are important.

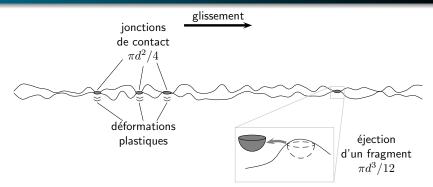
The plasticity index provides information on the risk of plastic deformation of a rough surface subjected to a load P:

$$\psi = \frac{P}{H} \left(\frac{\sigma}{R}\right)^{1/2}$$

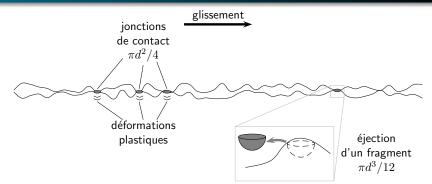
For  $\psi \le 0.6$  the deformation of the surface remains essentially elastic, while for values greater than unity, the deformation will be mainly plastic.

This index is used to predict the running-in periods, where a transition from the plastic domain to the elastic domain is observed.

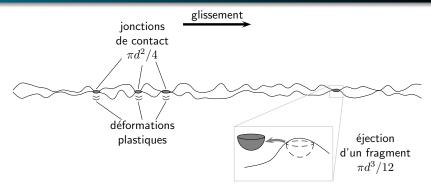




In the presence of adhesion ( $\alpha \le 5$ ) and if sliding is imposed, wear will occur.



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- Some junctions can break and lead to the formation of a material particle, which is usually transferred to the surface of the contacting body; this particle can then be ejected later as debris.
- Adhesive wear is directly related to the real contact area between the two bodies ≈ *P*/*H*.

# Archard Model

Simplified model<sup>[1]</sup> considering that the junctions have an average area of  $\pi d^2/4$  and that the fragments of material removed are hemispherical of volume  $\pi d^3/12$ .

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 $n = P/H/(\pi d^2/4)$ 

Considering a displacement  $\Delta l$ , Archard supposes that each junction is broken after a distance *d*. Thus, *N* junctions will have formed such that :

$$N = n\frac{\Delta l}{d} = 4\frac{P}{H}\frac{\Delta l}{\pi d^3}$$

J. F. Archard, Contact and rubbing of flat surfaces, Journal of Applied Physics, p 981-988 (1953).

If *K* is the probability for a junction to give rise to the transfer of a particle, the used volume is expressed by  $\Delta V = KN \pi d^3/12$ .

The worn volume per unit of slip is thus written :

 $\frac{\Delta V}{\Delta l} = \frac{KP}{3H}$ 

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We observe that the wear volume is proportional to the contact load as well as to the distance  $\Delta l$ , but inversely proportional to the hardness of the material used. *K* is called the *wear coefficient* and allows to compare various materials with each other for their resistance to wear. The higher the value of *K*, the greater the wear.

*K* represents the fraction of the contact junctions which will produce wear fragments.

- *K* = 1 : Every junction involved in the friction process produces a wear fragment.
- *K* = 0.1 : One tenth of the friction junctions produce wear fragments. For clean gold surfaces *K* is between 0.1 and 1. For clean-copper surfaces *K* is between 0.1 and 0.01. Clean gold surfaces wear about ten times more rapidly than clean copper surfaces.
- $K = 10^{-7}$ : One contact junction in ten million produces a wear fragment.

The following table presents values of wear coefficients for different combinations of materials in contact<sup>[1]</sup>. It should be noted that the presence of a lubricant allows the coefficient of wear to be reduced significantly.

Materials in contact	Wear coefficient <i>K</i>
Zinc on Zinc	$160 \cdot 10^{-3}$
Copper on Copper	$32 \cdot 10^{-3}$
Stainless steel on Stainless steel	$21 \cdot 10^{-3}$
Copper on Low carbon steel	$1,5 \cdot 10^{-3}$
Low carbon steel on Copper	$0, 5 \cdot 10^{-3}$
Bakelite on Bakelite	$0,02 \cdot 10^{-3}$

M. Shaw M, Friction and Wear, p 657-673, Addison Wesley, Reading MA (1966).

# Contents

## 1 Wear

Adhesive wear

#### Abrasive wear

- Chemical wear
- Experimental testing

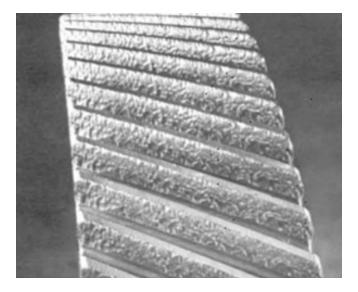
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- Fretting wear
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# Abrasive wear



# Abrasive wear description

- A hard surface penetrates a softer surface;
- the relative displacement of surfaces causes degradation.

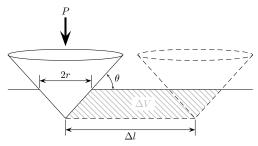
# Abrasive wear description

- A hard surface penetrates a softer surface;
- the relative displacement of surfaces causes degradation.
- two-bodies wear when the two materials in contact have very different hardnesses
- three-bodies wear when small, very hard particles are located between the two materials (introduced intentionally, as in the case of polishing, or generated by the first phases of two-body wear or tribochemical reaction).

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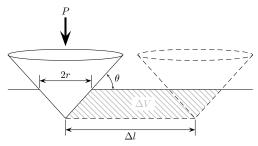
 $\rightarrow$  Wear manifests itself by the creation of plastic grooves on the surface of the worn material and the possible ejection of particles of material.

# Prediction of the wear volume $\Delta V$



Conical particle penetrating the softer material, displaced by a distance  $\Delta l$ .

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Conical particle penetrating the softer material, displaced by a distance  $\Delta l$ .

The volume of the groove created by the displacement of the conical particle is expressed by  $\Delta V = r^2 \tan \theta \Delta l$ , while the indentation pressure is directly related to the hardness *H* of the softer material and at the contact area :  $P = \pi r^2 H$ . We can therefore write :

$$\frac{\Delta V}{\Delta l} = \frac{\tan\theta}{\pi} \frac{P}{H}$$

# Prediction of the wear volume $\Delta V$

 $\frac{\Delta V}{\Delta 1} = K' \frac{P}{H}$ 

This law is globally of the same form as Archard's law except for the multiplicative coefficient. Compared to adhesive wear, wear coefficients depend essentially on the geometry of the abrasive particles and are more in the order of  $10^{-3}$  for three-bodies wear and  $10^{-2}$  for two-bodies wear. Two-bodies wear is therefore much more dangerous, with the third body playing the role of a solid lubricant.

# Contents

## 1 Wear

- Adhesive wear
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## Chemical wear

Experimental testing

## 2 Fretting

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- Fretting wear
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- Numerical analysis of fretting initiation and propagation
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4 Outlook : insight from numerical models

Contact degradations in the presence of a reactive environment are qualified as tribochemical wear. Damage is dominated by chemical reactions occurring between the contact surfaces and / or the external environment. These phenomena interact with mechanical stresses which can sometimes facilitate reactions.

If the medium is corrosive, we can also observe a phenomenon of tribocorrosion activated by contact stresses in the same way as damage by stress corrosion. The deterioration of wear is then greatly increased. The formation of oxides under these conditions (tribo-oxidation) can have either beneficial or harmful effects on the mechanical strength of the parts :

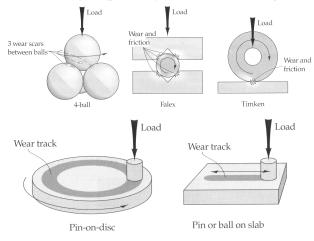
- the oxide layer formed can limit mechanical stresses, by lowering the coefficient of friction or by adaptated plasticity (plastic shakedown);
- if this layer fractures under the action of mechanical contact stresses, it is quickly eliminated. The action depends on whether the debris will effectively play the role of third body by lubricating the contact. If this is not the case (hard and abrasive debris) the damage can quickly become catastrophic.

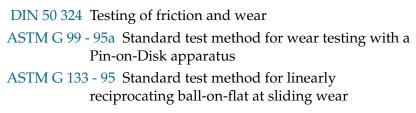
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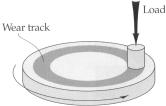
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- Abrasive wear
- Chemical wear
- Experimental testing
- 2 Fretting
  - Basics
  - Fretting wear
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Schematic illustration of some sample configurations used in simulation of dry or partially lubricated sliding contacts :







Pin-on-disc

# Comonly used parameters in the characterisation of tribological contacts

## **Operating parameters**

- Load
- Sliding speed
- Sliding distance

## Lubrication parameter

- Viscosity
- Flow rate
- Thermal conductivity
- Acidity
- Boiling point
- Solidification point

## **Material parameters**

- Hardness
- Toughness
- Melting point

### **Environmental parameters**

- Relative humidity
- Local air pressure
- Radiation level

## Pin on disc tribometer

- CSM company
- rotating specimen vs fixed pin
- environmental control chamber

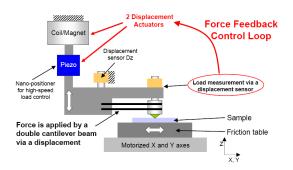


## High load UMT TriboLab

- CETR (now Brucker) company
- reciprocating sliding
- motorized 500 N normal load



Investigation of the phenomena occuring along the scratch deformation : cracking, spallation, delamination or bulking.

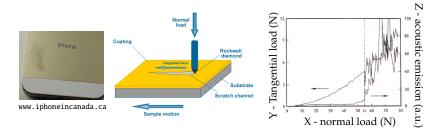


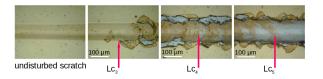
Quantify :

- Scratch resistance
- Adhesion of Coatings
- Friction Coefficient
- Viscoelastic properties
- Wear Testing
- Conventional Hardness

#### Scratch test to quantify coatings adhesion to substrate

 $\rightarrow$  scratching a surface with an indenter or ball to characterize the critical loads (*L*<sub>C</sub>) at which the coating failure failure occurs.





# Standard EN 1071-3

Three different scratching procedures :

- Progressive load scratch test (PLST)
  - loading rate 100 N/min
  - lateral displacement speed 10 mm/min
- Constant load scratch test (CLST)
  - loading rate 100 N/min
  - lateral displacement speed 10 mm/min
  - load step 1/5 of *L*<sub>C</sub>
- Multipass scratch test (MPST)
  - repeated scratching under a constant sub-critical load within the same scratch track
  - lateral displacement speed operating parameter same as for CLST
  - load 1/2 of  $L_C$
  - number of scratches until failure







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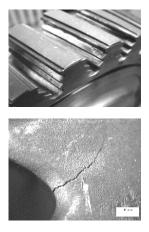
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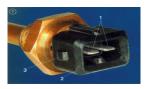
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4 Outlook : insight from numerical models

# Fretting problems



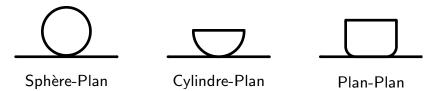
ex : turbine blade/disk contact (cracking)





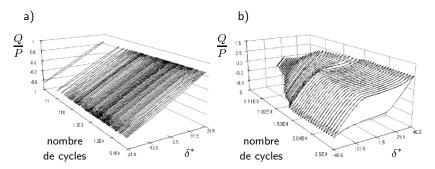
ex : electrical contact (wear)

Very complex real contact geometries  $\rightarrow$  simplification of the geometry for laboratory testing.

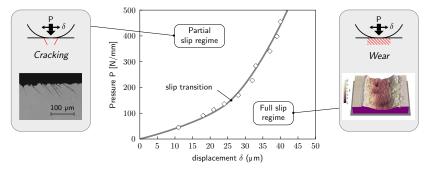


- The sphere-plane and plane-plane configurations are often used to study the kinetics of fretting wear;
- rather, the cylinder-plane configuration is used to study fretting cracking.

The curve  $Q(\delta)$  adopts two characteristic shapes depending on the loading parameters :



# Running condition fretting map



- At low displacement amplitudes and high normal load : partial slip regime. Cracking is mainly observed.
- At large displacement amplitudes : full sliding regime.
   Wear is the dominant degradation mechanism.

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#### 2 Fretting

Basics

#### Fretting wear

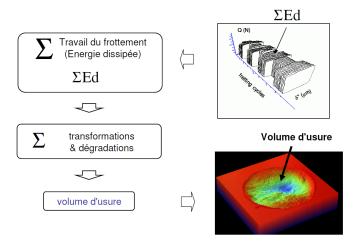
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3 Modelling fretting taking into account the contact gradient

4 Outlook : insight from numerical models

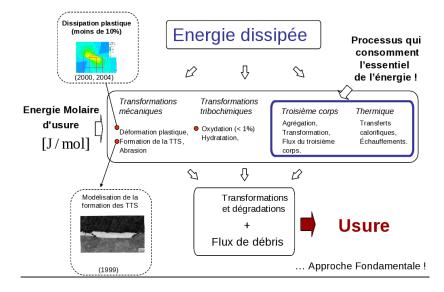
# Fretting wear : dissipated energy

Fretting usure : approche energétique<sup>[1]</sup>



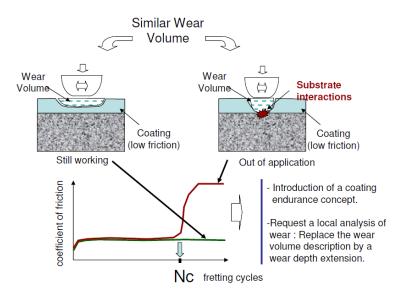
S. Fouvry, C. Paulin and S. Deyber, Tribology International, 42(3):461-474 (2009).

#### Fretting wear : thermodynamic approach

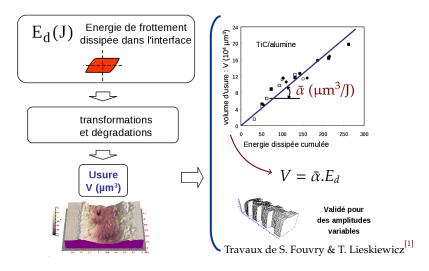


# Fretting wear : local approach

Fretting Usure : Approche Locale

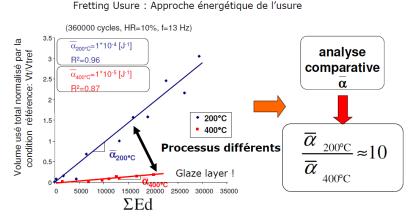


# Friction energy capacity to predict surface coating endurance



T. Liskiewicz and S. Fouvry, Tribology international, 38(1):69-79 (2005).

#### Example of Comparative analysis



Ex. Contact Inox/Inox

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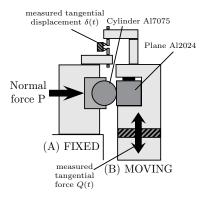
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- Basics
- Fretting wear
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# Example of experimental fretting tests

#### Fretting wear configuration

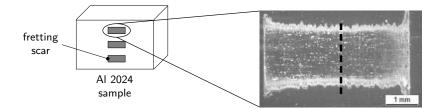
- Flat vs. Cylinder contact
- AA2024 damage tolerant aerospace alloy
- Partial slip condition
- Measure of P, Q(t) and  $\delta(t)$  during test
- number of cycles  $0 < N < 4.10^{6}$



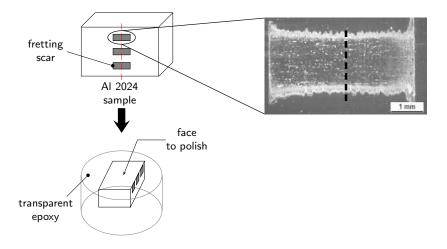
Fretting experiments carried out at Ecole Centrale de Lyon



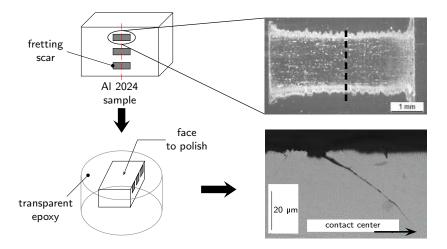
# Fretting damage investigation



# Fretting damage investigation

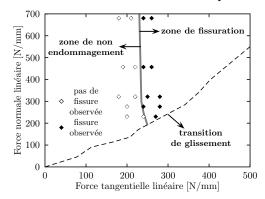


# Fretting damage investigation



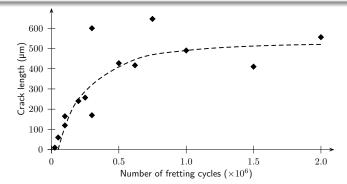
#### Fretting crack initiation boundary

- crack initiation boundary at 50k cycles for a dry contact aluminium-aluminium
- no apparent role of the contact pressure
- tangential force crack initiation threshold  $Q_c = 240$  N/mm. The initiation in this case seems driven by the surface shear.



#### Experimental conditions

- Fretting loading :  $p_0 = 325$  MPa,  $\delta = 20 \ \mu m$ ;
- Destructive characterisation : 1 experimental point ⇔ 1 test.



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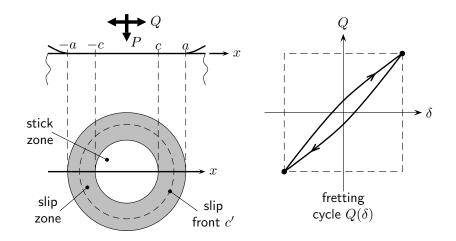
#### 1 Wear

- Adhesive wear
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#### 2 Fretting

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- Fretting wear
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# Fretting contact in the partial slip regime



# Theory of superposition

Formulation by McEwen<sup>[1]</sup> and applied to contact loading like fretting by Johnson<sup>[2]</sup>, later extended by Hills<sup>[3]</sup>:

Analytical stress distribution for sliding contact

 $\sigma = \sigma^{P} + \sigma^{Q}$  (Hertz and Coulomb)

E. McEwen, Philosophical Magazine, 40:454-459 (1949).

K. L. Johnson, Contacts Mechanics, Cambridge University Press, (1985).

D. Hills, D. Nowell and A. Sackfield, Mechanics of Elastic Contacts, (1993).

# Theory of superposition

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Analytical stress distribution for sliding contact

 Partial slip stress distribution (proposed by Cattanéo and Mindlin)

$$\begin{split} & \underline{\sigma}(\underline{X}\,,c') = \underline{\sigma}^P + \underline{\sigma}^Q(\underline{X}\,,c) + \underline{\sigma}^Q(\underline{X}\,,a) - 2\underline{\sigma}^Q(\underline{X}\,,c') \text{ load} \\ & \underline{\sigma}(\underline{X}\,,c') = \underline{\sigma}^P - \underline{\sigma}^Q(\underline{X}\,,c) - \underline{\sigma}^Q(\underline{X}\,,a) + 2\underline{\sigma}^Q(\underline{X}\,,c') \text{ unload} \end{split}$$

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# Theory of superposition

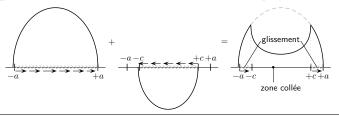
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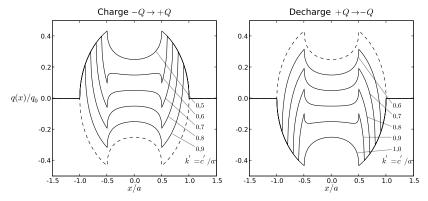
Example at the surface at the end of the fretting cycle (c = c') :



- E. McEwen, Philosophical Magazine, 40:454-459 (1949).
- K. L. Johnson, Contacts Mechanics, Cambridge University Press, (1985).
- D. Hills, D. Nowell and A. Sackfield, Mechanics of Elastic Contacts, (1993).

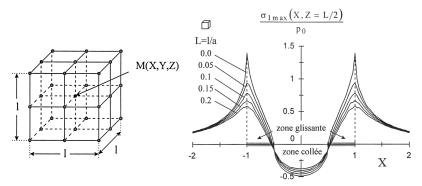
#### Shear stress during the fretting cycle

Evolution of the surface shear during the fretting cycle in partial slip predicted by the theory of Mindlin ( $\mu = 0, 5$  and k = c/a = 0, 5):



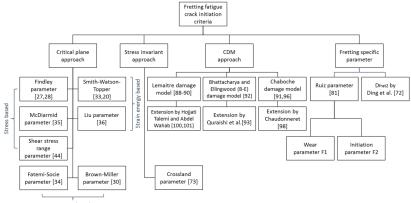
#### Multiaxial fatigue criterion for crack initiation

- From the stress fields and the elastic contants, the strain field can be derived (or computed numerically including plasticity effects).
- from *σ* and *ε* fields, a multiaxial fatigue criterion can be used to predict the initiation location and life.
- Due to the severe stress/strain gradients, a spatial averaging procedure is needed.



#### A fretting criterion classification

Review of the various criteria used to predict fretting crack initation<sup>[1]</sup>



Strain based

Ex for the Findley Parameter :

$$FP = \frac{\tau_{max}}{2} + k_1 \sigma_{max}^n \quad \text{and initiation for} \quad \frac{\tau_{max}}{2} + k_1 \sigma_{max}^n = \tau_f'(2N_i)^{b'}$$

N. A. Bhatti and M. A. Wahab, Tribological Transactions, in press.

#### Contents

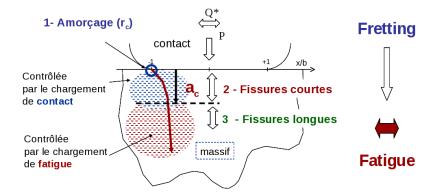
#### 1 Wear

- Adhesive wear
- Abrasive wear
- Chemical wear
- Experimental testing

#### 2 Fretting

- Basics
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- Experimental determination of fretting crack initiation
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- Fretting fatigue
- 3 Modelling fretting taking into account the contact gradient
- 4 Outlook : insight from numerical models

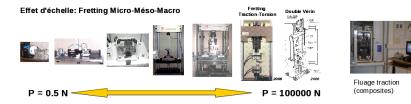
# Fretting fatigue cracking



« Comment prédire la cinétique de fissuration en Fretting ? »

# Fretting at LTDS, Lyon (France)

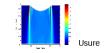
#### Plateforme expérimentale Fretting & Fatigue



#### Plateforme numérique Fretting & Fatigue

- Centre FEM ABAQUS (macro 2D // Visco-Elasto-Plastique )
- Critères multiaxiaux de fatigue
- Propagation des fissures (Crack-Box, Fonctions de poids & Dislocations)
- Codes d'usure remaillage des surfaces (Wear-Box)

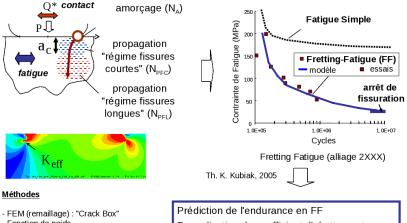




Fissure

# From fretting-fatigue maps to life predictions

$$N_T = N_A + N_{PFC} + N_{PFL}$$



- Fonction de poids
- Distribution de Dislocations

Prédiction de l'endurance en FF Formalisation du coefficient d'abattement Fretting-Fatigue / Fatigue

S. Fouvry and K. Kubiak, International Journal of Fatigue, 31(2):250-262 (2009).

S. Fouvry and K. Kubiak, Wear, 267(12) :2186-2199 (2009).

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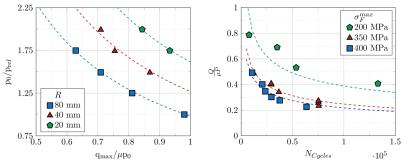
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Fretting problems suffer from severe **stress gradients** located at the transition from stick and slip zones, which are known to imped classical fatigue approaches.

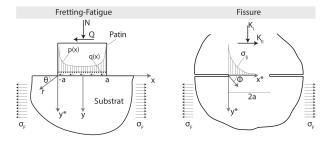
For instance :

- varying the contact radius in fretting tests;
- varying the nominal fatigue stress in fretting fatigue tests gives different crack initiation boundaries.



# Crack analogue approach

For a **pure stick contact**, the stress fields at the root of the contact notch are similar to those present at a crack tip.



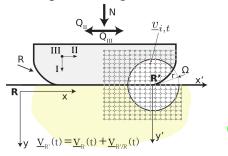
Using the contact stress field solution :

• 
$$\sigma_{yy}(x \to a, y = 0) = \frac{N}{\pi \sqrt{2ar}} = \frac{K_I}{\sqrt{\pi a}}$$
 thus  $K_I = -\frac{N}{\sqrt{\pi a}}$   
• and similarly for mode II :  $K_{II} = \frac{Q}{\sqrt{\pi a}}$ 

The stress intensity factors  $K_I$  and  $K_{II}$  depends on the applied load but also on *a*, directly **integrating the contact gradient**.

#### Reduced Order Modelling of partial slip contact

**Proper Orthogonal Decomposition** of the velocity field (computed using finite elements) using 5 modes.



$$\underline{v}(\underline{P},t)_{R'} = \underline{v}_{e}(\underline{P},t)_{R'} + \underline{v}_{c}(\underline{P},t)_{R'}$$
linear part
stick contact)
non linear part
(slip contact)

$$\underline{v}_{e}(\underline{P},t)_{R'} \simeq \dot{I}_{I}(t)\underline{\phi}_{I}(\underline{P}) + \dot{I}_{II}(t)\underline{\phi}_{II}(\underline{P}) + \dot{I}_{III}(t)\underline{\phi}_{III}(\underline{P})$$

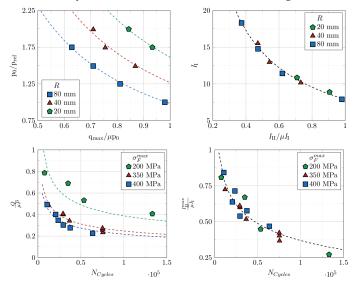
$$\underline{v}_{c}(\underline{P},t)_{R'} \simeq \dot{I}_{II}^{c}(t)\underline{\phi}_{II}^{c}(\underline{P}) + \dot{I}_{III}^{c}(t)\underline{\phi}_{III}^{c}(\underline{P})$$

The 3D contact is approximated using 5 dof :  $\dot{I}_{I}$ ,  $\dot{I}_{II}$ ,  $\dot{I}_{III}$ ,  $\dot{I}_{II}^{c}$ ,  $\dot{I}_{II}^{c}$ 

G. Rousseau, Modélisation de la durée de vie en fretting-fatigue sous chargements complexes, PhD thesis, Univ Paris-Saclay (2020).

# A unified model for fretting fatigue predictions

The model is able to predict fretting and fretting fatigue life under many different conditions and stress gradients !



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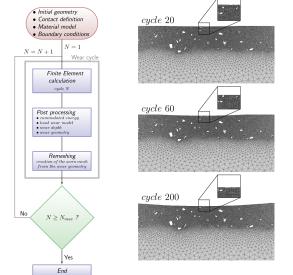
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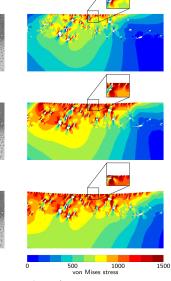
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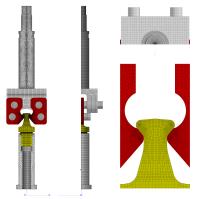
#### Wear models





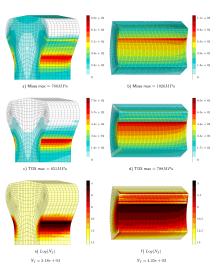
H. Proudhon *et al.*, Experimental and numerical wear studies of porous reactive plasma sprayed Ti-6Al-4V/TiN composite coating. *Wear*, (311) :159-166, 2014.

# Fretting calculations on a technological specimen

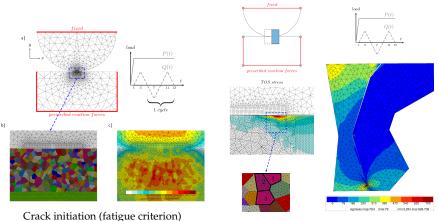


blade-disk technological testing at Safranelastoplastic coating with workhardening

• Fields transfer



# Crack propagation model (polycristal)



Crack propagation (crystal plasticity)

PhD thesis L. Sun (2012), H. Proudhon et al., Matériaux et Techniques, (101):203, 2013.